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## **An FDEM study of particle breakage under rotational point loading**

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## Nomenclature

$C_n^m$	Spherical Harmonic coefficients of degree $n$ and order $m$
$D$	damage variable for linear damage evolution
$D'$	damage parameter
$d$	particle size
$E$	elastic modulus
$E_i, E_k, E_D, E_E, E_S, E_W$ and $E_F$	internal, kinetic, damage, elastic, strain, input and friction energies
$F_{\max}$	particle breakage force
$\mathbf{F}_e$ and $\mathbf{F}_i$	external and internal forces
$G_C$	critical fracture energy
$G_{n,C}$ and $G_{shear}$	critical fracture energies for failure in the normal and shear directions
$\mathbf{K}$	elasticity matrix
$k_1, k_2, k_M$ and $k_g$	maximum, minimum, average and Gaussian curvature values
$k_C$ and $k_{com}$	curvedness and combined curvedness
$k_n, k_s$ and $k_t$	CIE stiffness along normal, the first and second shear directions
$L_{min}$	minimum mesh size
$\mathbf{M}$	diagonal mass matrix
$N_{\max}, S_{\max}$ and $T_{\max}$	failure stresses towards normal, the first and second shear directions
$R^*$	relative contact radius of the Hertzian contact model
$R_1$ and $R_2$	radius of two contacted spheres
$r_i(\theta, \varphi)$	distance between the mass center and $i$ -th point on particle surface
$T$	thickness of cohesive elements
$t$	time
$u, \dot{u}$ and $\ddot{u}$	displacement, velocity and acceleration
$\ddot{\mathbf{u}}$	acceleration vector
$V$	particle volume
$Y_n^m(\theta, \varphi)$	Spherical Harmonic functions of latitudinal coordinate $\theta$ and longitudinal coordinate $\varphi$
$\boldsymbol{\sigma}$	nominal traction stress vector
$\boldsymbol{\sigma}^c$	stress from overall constitutive equation of particle
$\boldsymbol{\sigma}^u$	undamaged stress
$\sigma_f$	particle breakage or failure stress
$\sigma_n, \sigma_s$ and $\sigma_t$	stresses towards normal, the first and second shear directions
$\sigma_{eff}^0$	effective traction stress
$\boldsymbol{\delta}$	displacement matrix
$\delta_n, \delta_s$ and $\delta_t$	displacements towards normal, the first and second shear directions
$\delta_m^0$ and $\delta_m^{sep}$	displacements for crack initiation and complete failure of CIEs
$\varepsilon$	strain
$\dot{\varepsilon}^{el}$	elastic strain rate
$\rho$	density of bulk material
$\eta$	material parameter for the Benzeggagh-Kenane criterion
$\chi$	energy-based mode mixed ratio
$\mu$	friction coefficient
$\Delta t$	time step

## Abbreviations

CIE	Cohesive interface element
LBS	Leighton Buzzard sand
XCT	X-Ray computed tomography

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8

### 9 **Abstract**

10 The most commonly adopted method to test the strength of single sand particles is based  
11 on platen experiments. This setup tends to align the loading direction towards the  
12 particle minimum axis and provide an upper limit for the breakage stress. This paper  
13 numerically bypasses such limitation by using a combined finite and discrete element  
14 method (FDEM). FDEM was first validated via a mesh size analysis of a spherical  
15 particle and calibrated by in-situ experimental compressions of the single quartz sand  
16 particle, where the particle shape was obtained by X-ray micro-computed tomography  
17 (XCT) and then imported into the numerical model. Systematic point loading tests were  
18 recreated to explore the role of the curvature at contacting points on the breakage  
19 behaviour. The simulations allow to probe the same non-spherical particles, i.e.,  
20 realistic quartz sand and ellipsoid particles, with multiple measurements highlighting  
21 the importance of the loading direction, which was inaccessible experimentally. Results

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1 show that FDEM can capture not only the crack initiation but also fracture patterns,  
2 while taking into account realistic shapes. It is found that the distance between two  
3 contact points and their combined curvedness reflecting the particle morphology are the  
4 two major factors governing fracture patterns and stresses. When loading is roughly  
5 parallel to the minimum principal dimension of particles, the obtained breakage stress  
6 and the number of fragments approach the upper limits.

7 **Keywords:** Particle breakage; point loading; FDEM; computed tomography; crack  
8 initiation; crack evolution; fracture patterns

#### 9 **Highlights**

- 10 i) An FDEM benchmarked with XCT was used to simulate particle breakage  
11 focusing on shape effects.
- 12 ii) Rotational point load simulations probed the same non-spherical particle  
13 breakage.
- 14 iii) Breakage stress and patterns are governed by loading distances and the  
15 curvature of contact points.
- 16 iv) Maximum stored elastic energy during deformation can be an effective breakage  
17 criterion.

18

## 1 **1. Introduction**

2       The micromechanical understanding of particle breakage targeting particle  
3 morphology, size and packing structure, bulk material properties and loading conditions  
4 is sought-after in an extensive range of applications including granular materials, soil  
5 mechanics, geology, and across the broader field of engineering [1-3]. Due to the  
6 difficulty in quantifying particle morphology features, particle size is usually taken as  
7 a first-order approximation for correlating breakage behaviours. There are two types of  
8 models considering size effects in the strength of material and structure design:  
9 deterministic size effect based on stress redistribution and released energy, e.g., un-  
10 notched size effect law [4-5]; and statistical size effect considering material defects,  
11 e.g., using Weibull distribution [6-7].

12       Stress-induced single particle breakage associated with its morphology features is  
13 of fundamental importance in geotechnical applications including blast loading, pile  
14 driving and large rockfill dams. Methods to study single particle strength or breakage  
15 can be classified into three categories: experimental [7-8], analytical [9-11] and  
16 numerical [12-14] approaches.

17       In the experimental approach, due to the small size of sand particles, nearly all 1D  
18 compression tests to measure individual particle strength through flat platen loading,  
19 where a single particle is compressed between two flat rigid platens. Displacement and  
20 force at the loading platen on the particle are simultaneously recorded. Point loading,  
21 widely applied for large-sized rock blocks [15], is a more realistic testing method, since  
22 most contacts inside granular assemblies are point-to-point rather than point-to-flat-

1 face contacts. Based on point load tests, most analytical methods [16-17] are conducted  
2 with the assumption of hard-soft contacts between load caps and spherical particles. In  
3 fact, the mechanical properties of compressing platens govern the breakage behaviour  
4 of the quartz sphere via the ratio between contacting radius ( $a$ ) and sphere radius ( $R$ ). For  
5 low ( $a/R < 0.3$ ), intermediate ( $0.3 < a/R < 0.65$ ) or high ( $a/R > 0.65$ ) contact radius, the  
6 corresponding particle breakage occurs just outside the contact perimeter, near the  
7 sphere centre, or along the particle equator due to the shift of maximum tensile stress  
8 [18]. Apparently, the location of fracture initiation is hard to detect experimentally even  
9 though necessary to bridge analytical solution and experiments. From the analytical  
10 solution of Russell and Wood [19], the breakage occurs just below contacting points  
11 and is driven by shear stresses, which have been qualitatively reflected by many small  
12 fragments below contacting area. This has been captured by the X-ray micro computed  
13 tomography (CT) images [20]. Zhu and Zhao [21] simulated the compression of a single  
14 sphere particle using peridynamics and found the fracture pattern to be similar to the  
15 experimental breakage behaviour of an LBS particle with the so-called good roundness  
16 contact point in Zhao et al. [20]. However, such peridynamics simulations did not  
17 reproduce the typically observed small shear-induced fragments near contact areas.

18 When platen loading tests are applied for irregular-shaped particles, the most used  
19 equation for calculating irregular particle strength is taken from the analytical and  
20 experimental approach of point load tests in [16]:  $\sigma_f = F/d^2$ , where  $F$  is the breakage  
21 force,  $\sigma_f$  is the breakage stress, and  $d$  is the distance between the two contact points at  
22 the particle surface in point load tests. It should be highlighted that if the particle,

1 particularly of non-convex shape, is laid on a bottom platen, at least three contact points  
2 would theoretically develop, either macroscopic or at asperity level contact points, on  
3 the bottom platen and more than one contact point with the upper platen, although  
4 during the consequent loading process of the compressed particle the contact points  
5 may vary. Therefore, occasionally the extremely non-spherical particle does not endure  
6 purely compressive loading but also structural bending, which may hinder the extended  
7 use of the simplified equation in the analysis.

8       Regardless of the difficulties in quantitatively correlating particle morphology, in  
9 terms of shape parameters and curvature of contact points, there are experimental  
10 investigations on single particle breakage behaviour in 1D compression that can be  
11 highlighted. Zhao et al. [20] found the stress concentration to occur at contact points  
12 with high curvature via X-ray CT, which was also observed by Wang and Coop [8].  
13 Cavarretta et al. [22] discussed the relevance of roundness from 2D curvature to particle  
14 crushing strength and concluded that rounded particles can endure high loaded stress  
15 due to the increased possibility of larger contact radii. Brzesowsky et al. [23] polished  
16 the bottom of crushing irregular particles to discover the relation between the curvature  
17 of only one contact point touching the top platen and breakage force for simplification.  
18 However, the curvature values in their experiments were based on one or several  
19 randomly selected 2D particle projections, which may not be sufficiently representative  
20 in 3D. Therefore, they found no obvious quantitative relation between particle breakage  
21 stress and curvature index.

22       A complementary tool to investigate particle breakage behaviour can be based in

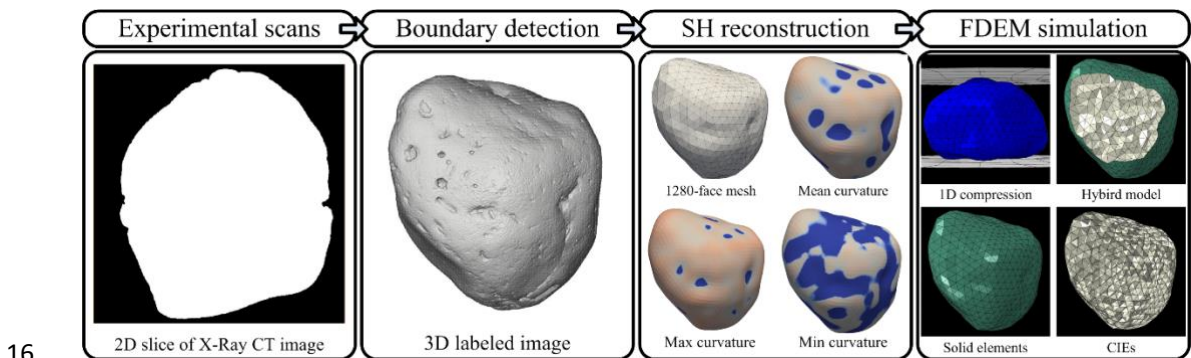
1 computational mechanics. Discrete element method (DEM) is currently the most often  
2 adopted in the context of geotechnical engineering for sand particles. Two main  
3 approaches for handling particle breakage can be identified: one is based on bonded  
4 clusters, where a particle is composed of bonded small sphere elements [12-13, 24-25];  
5 and another approach replaces a ‘broken’ particle with a few smaller spherical particles  
6 [26-28]. Unfortunately, both approaches lack rigorous theoretical basis of fracture  
7 mechanics at the particle scale. In recent years, the development of 3D X-ray micro-  
8 tomography and laser techniques for accurate reconstruction of sand particle surfaces  
9 opened the possibility for simulating realistic particle morphology with computational  
10 models, such as the granular element method (GEM) [29] and level set DEM (LS-DEM)  
11 [30-31]. The finite element method (FEM) with surface meshes is more suitable for  
12 depicting realistic particle morphology when compared with other DEM-based  
13 computational tools [32]. Druckrey and Alshibili [33] implemented the extended finite  
14 element method (XFEM) to simulate the 1D compression of single quartz sand particles  
15 based on in-situ X-ray synchrotron images. However, due to the features of the enriched  
16 elements used to simulate smeared fracture and difficulties in handling differences in  
17 density within the particle, only one flat fracture surface connecting two loading areas  
18 could develop, which may be insufficient for capturing the fracture patterns found in  
19 experiments and including four typical modes, namely major splitting, explosive,  
20 chipping and mixed modes [8].

21 In this study, a combined finite and discrete element method (FDEM) approach is  
22 adopted to simulate particle breakage behaviour, via multi-directional two-point

1 loading, focusing on the particle shape effects. The simulations of rotational point  
 2 loading provide a unique advantage since it allows probing the same non-spherical  
 3 particle shapes with multiple breakage measurements assessing the role of loading  
 4 direction, which is inaccessible in experiments. The paper is organised as follows. In  
 5 Section 2, the FDEM approach is introduced and benchmarked with XCT-monitored  
 6 experiments. In Sections 3 and 4, spherical quartz particles are simulated with variant  
 7 relative contact radius, to bridge breakage behaviour and curvature values of contact  
 8 points. Rotational loading simulations of the XCT scanned particle and ellipsoid are  
 9 also conducted and analysed, revealing that particle breakage behaviour, regarding  
 10 breakage stress and fracture patterns, is governed by both their loading distance and the  
 11 curvature of contact points. Further analyses show that the maximum stored strain  
 12 energy during deformation can be an effective breakage criterion to determine brittle  
 13 particle strength. Finally, the main conclusions are summarised in Section 5.

14

15 **2. Combined finite and discrete element method**



17 Fig. 1 From experimental in-situ X-ray computed tomography to FDEM simulation of particle  
 18 breakage behaviour using platen loading

1 FDEM combines FEM with contact detection and interaction to simulate not only the  
2 continuum behaviour within the particle but also the initiation and development of  
3 internal cracks and the interactions between fragments. Abaqus/Explicit [34] is herein  
4 adopted for developing the FDEM simulations given its ability to deal with a large  
5 number of contacts between discontinuous elements undergoing large deformation.

## 6 **2.1 Image processing and mesh generation**

7 Fig. 1 shows a typical workflow for preparing our FDEM simulations of single  
8 particle crushing considering realistic particle shape. There are four steps: (1) the CT-  
9 monitored experiment for an LBS particle in [20] offering a series of X-Ray raw CT  
10 images during platen loading; (2) extraction of 3D voxel-based particle shape using  
11 image processing; (3) reconstruction of particle morphology with Spherical Harmonic  
12 analyses; and (4) discretization into 3D tetrahedron meshes for FDEM simulation.

13 The 3D CT data can be visualised as a stack of 2D images, each with a cross-  
14 section of the scanned sample with the thickness of one voxel of high resolution of 3.3  
15  $\mu\text{m}$ . The image processing and scanning from CT-monitored sets of 2D slice images to  
16 reconstruct the particle mesh is done following the procedure described in [20]. The  
17 main steps include the use of a 2D median filter [35] to reduce the noise on every slice,  
18 then followed by binarization of the filtered image for detection of the LBS particle  
19 boundaries from distinct material phases. The labelled CT images are smoothed using  
20 spherical harmonics (SH) in polar coordinates by decomposing a spherical scalar  
21 function as follows [36]:

$$r_i(\theta, \varphi) = \sum_{n=0}^{\infty} \sum_{m=-n}^n c_n^m Y_n^m(\theta, \varphi) , \quad (1)$$

where  $i$  denotes  $i$ -th points on CT-based particle surface;  $r$  is the distance between surface points and particle mass centre;  $Y_n^m(\theta, \varphi)$  and  $c_n^m$  are the SH functions and coefficients at degree  $n$  and order  $m$  defined in [37], respectively; and  $\theta \in [0, \pi]$  and  $\varphi \in [0, 2\pi]$  are the latitudinal and longitudinal coordinates.

To perform rotational point load tests for the same particle, of which only vertices of triangular meshes contact with load point ends, the selected surface point ( $r_i(\theta, \varphi)$ ) have a corresponding point ( $r_j(\pi - \theta, \varphi + \pi)$ ) and the mass centre is on the line connecting them, as in Fig. 2. Finally, the generated surface mesh is converted into tetrahedron-composed solid domain [38].

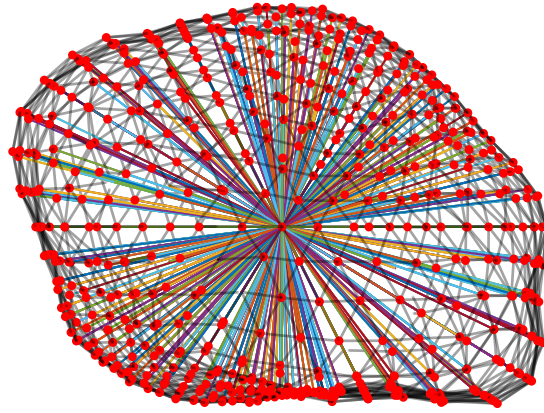


Fig. 2 The ‘diameters’ connecting the centre-of-mass and the vertices on particle surface

## 2.2 Model description

A particle is herein discretised in the context of FDEM by 4-node solid tetrahedron elements connected by zero thickness cohesive interface elements (CIEs in Fig. 1). The

1 tetrahedron elements are adopted for their compatibility with the triangular surface  
 2 mesh reconstructed from CT images. The interface elements connect the adjacent solid  
 3 elements until fracture initiation conditions are met. Fragments are generated when a  
 4 throughout fracture plane develops. To allow for the multiple possibilities of fracture  
 5 propagation, the tetrahedron elements need to be small enough. Following standard  
 6 FEM principles, the nodal matrix for a moving element is governed by the following  
 7 equilibrium equation:

$$8 \quad \mathbf{M} \cdot \ddot{\mathbf{u}} + \mathbf{F}_i - \mathbf{F}_e = \mathbf{0} , \quad (2)$$

9 where  $\ddot{\mathbf{u}}$  is the acceleration vector,  $\mathbf{M}$  is the diagonal mass matrix and  $\mathbf{F}_i$  and  $\mathbf{F}_e$  are  
 10 internal and external forces.  $\mathbf{F}_i$  is composed of elastic force from isotropic solid  
 11 tetrahedron elements and intact CIEs;  $\mathbf{F}_e$  is resulted from the external contacts between  
 12 particle and platens or between newly generated fragments.

13 A penalty formulation is used for contact with a tangential friction coefficient  
 14 ( $\mu=0.5$ ) between fractured solid elements, and between solid elements and rigid platens .

15 An explicit solver is adopted to obtain the solution via central difference integration  
 16 framework, due to its higher efficiency in dealing with large number of contact and  
 17 deformation when compared to implicit solvers. During the explicit integration scheme,

$$18 \quad \dot{u}_{n+1/2} = \dot{u}_{n-1/2} + \frac{\Delta t_{n+1} + \Delta t_n}{2} \ddot{u}_n \quad (3)$$

$$19 \quad u_{n+1} = u_n + \Delta t_{n+1} \dot{u}_{n+1/2} \quad (4)$$

$$20 \quad \Delta t = \sqrt{\frac{\rho}{E}} L_{\min} \quad (5)$$

21 where  $u$  is a freedom degree,  $n$  means  $n$ -th time step or increment, and  $\Delta t$  denotes time

1 step;  $\rho$  is the density of bulk material,  $E$  is the elastic modulus, and  $L_{\min}$  is the minimum  
 2 length of mesh size. For the balance between computation efficiency and accuracy,  
 3 mesh size should be selected carefully and will be discussed hereafter.

### 4 **2.3 Damage model**

5 Fracture or breakage is herein simulated by the failure of geometric zero-thickness  
 6 cohesive elements using traction-separation damage laws [39-41]. The relationship  
 7 between displacement and traction force at the cohesive zone follows the bilinear  
 8 function, as stated in Fig. 3:

$$9 \quad \boldsymbol{\sigma} = \begin{pmatrix} \sigma_n \\ \sigma_s \\ \sigma_t \end{pmatrix} = \mathbf{K} \boldsymbol{\delta} = \begin{pmatrix} k_{nn} & k_{ns} & k_{nt} \\ k_{sn} & k_{ss} & k_{st} \\ k_{tn} & k_{ts} & k_{tt} \end{pmatrix} \begin{pmatrix} \delta_n \\ \delta_s \\ \delta_t \end{pmatrix}, \quad (6)$$

10 where  $\boldsymbol{\sigma}$  is the traction stress vector,  $\mathbf{K}$  is the elasticity matrix,  $\boldsymbol{\delta}$  is the relative  
 11 displacement matrix, and  $n$ ,  $s$  and  $t$  denote the normal and two shear directions. Off-  
 12 diagonal components of the elastic matrix  $\mathbf{K}$  are set to zero. The relative displacements  
 13 are defined as:

$$14 \quad \delta_{n,s,t} = \varepsilon_{n,s,t} T, \quad (7)$$

15 where  $\varepsilon$  is the strain along the three directions for the CIEs and  $T$  is the fictitious  
 16 thickness herein set as 1 for the equality of nominal strain to its corresponding  
 17 displacement [42-43].

18 The constitutive response of the CIEs covers two main stages, i.e., before and after  
 19 breakage initiation. The maximum nominal stress criterion is applied to identify the  
 20 breakage initiation in quartz sands as suggested in [33, 38]:

$$MAX \left\{ \frac{\langle \sigma_n \rangle}{N_{\max}}, \frac{\sigma_s}{S_{\max}}, \frac{\sigma_t}{T_{\max}} \right\} = 1$$

$$\langle \sigma_n \rangle = \begin{cases} \sigma_n, & \text{for } \sigma_n \geq 0 \\ 0, & \text{for } \sigma_n \leq 0 \end{cases}, \quad (8)$$

where  $N_{\max}$ ,  $S_{\max}$  and  $T_{\max}$  are the failure stress of CIEs. Compressive stress is set to 0 to evaluate crack initiation criterion since compression is not considered as a source of breakage at micro-scale.

After crack initiation, contact stiffness reduces according to partial interface damage, and the stress-displacement relationship is updated by (Fig. 3):

$$\begin{aligned} \sigma_n &= \begin{cases} k_n \delta_n & \text{for } \delta_n \geq 0 \\ k_n^0 \delta_n & \text{for } \delta_n \leq 0 \end{cases} \\ \sigma_s &= k_s \delta_s \\ \sigma_t &= k_t \delta_t \end{aligned}, \quad (9)$$

where  $k_{n,s,t} = (1-D)k_{n,s,t}^0$ .  $D$  is the damage variable for linear damage evolution defined as:

$$\begin{aligned} D &= \frac{\delta_m^{sep} (\delta_m^{\max} - \delta_m^0)}{\delta_m^{\max} (\delta_m^{sep} - \delta_m^0)} \\ \delta_m^{sep} &= \frac{2G_C}{\sigma_{eff}^0} \end{aligned}, \quad (10)$$

where  $\delta_m^{sep}$  and  $\delta_m^0$  are the corresponding displacements for complete failure and crack initiation, and  $\delta_m^{\max}$  is the maximum displacement registered during the loading history,  $\sigma_{eff}^0$  ( $= \sqrt{\langle \sigma_n \rangle^2 + \sigma_s^2 + \sigma_t^2}$ ) is the relatively effective traction stress at crack initiation, and  $G_C$  is the Griffith fracture energy for crack evolution.

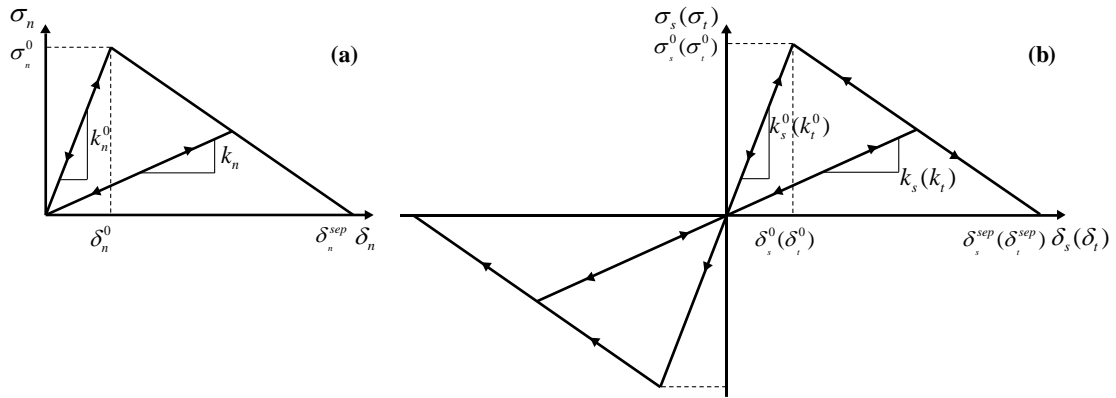
1 The Benzeggagh-Kenane criterion [44] is used to consider the mixed mode  
 2 opening:

$$G_C = G_{n,C} + (G_{s,C} - G_{n,C}) \left( \frac{G_{shear}}{G_T} \right)^\eta$$

$$G_{shear} = G_s + G_t$$

$$G_T = G_n + G_s + G_t$$
(11)

3  
 4 where  $G_{n,C}$  and  $G_{shear}$  are the relatively critical energies for pure mode I (tension) and  
 5 combination of mode II and mode III (shear),  $\eta$  is a material parameter, and  $G_T$  is the  
 6 mixed fracture energy for CIEs.



7  
 8 Fig. 3 Bilinear constitutive relations of CIEs: (a) Normal component; (b) Shear  
 9 components

10 An energy-based mode mixed ratio is defined to detect the dominant type of  
 11 damage during breakage:

$$\chi = 1 - \frac{G_n'}{G_n' + G_s' + G_t'}$$
(12)

12  
 13 where  $G_n'$ ,  $G_s'$  and  $G_t'$  are the fracture energy done by tractions along corresponding  
 14 displacements of normal, first and second shear directions, respectively. Accordingly,  $\chi$

1 is set to -1 before crack initiation.  $\chi = 0$ ,  $\chi = 1$  and  $0 < \chi < 1$  denote pure tension damage,  
 2 pure shear damage and mixed damage modes, respectively. When defined during crack  
 3 evolution or at crack initiation,  $\chi$  varies or remains constant with time at an integration  
 4 point.

5 Table. 1 summarises the main parameters obtained from the previous  
 6 publications for quartz. These include the measurable physical parameters of quartz  
 7 sand particle including density ( $\rho$ ), elastic modulus ( $E$ ), Poisson's ratio ( $\nu$ ), tensile  
 8 strength ( $N_{max}$ ) and shear strength ( $S_{max}$  and  $T_{max}$ ) (Alshibili et al., 2013) and CIE stiffness  
 9 ( $k_n$ ,  $k_s$  and  $k_t$ ), fracture energy ( $G_n$ ,  $G_s$  and  $G_t$ ) and material parameter ( $\eta$ ) and their  
 10 calibration process in FDEM [45-47]. The selection of adequate mesh size is key to  
 11 accurately represent the fracture initiation and fracture energy; the element size should  
 12 be less than the length of fracture propagation zone (FPZ). A rough estimation of the  
 13 length of FZP was done following [45]. Then according to [71-72], a mesh size  
 14 sensitivity study can be more targeted for the accuracy representation of fracture path.

15

	Parameter	Value
Solid elements	Density, $\rho$ (kg/m <sup>3</sup> )	2650
	Elastic modulus, $E$ (GPa)	94.4
	Poisson's ratio, $\nu$	0.118
Cohesive interface elements (CIE)	Normal stiffness, $k_n$ (N/m <sup>3</sup> )	$9.44 \times 10^{13}$
	First shear stiffness, $k_s$ (N/m <sup>3</sup> )	$4.22 \times 10^{13}$
	Second shear stiffness of, $k_t$ (N/m <sup>3</sup> )	$4.22 \times 10^{13}$
	Tensile strength, $N_{max}$ (MPa)	25.3
	First shear strength, $S_{max}$ (MPa)	12.6
	Second shear strength, $T_{max}$ (MPa)	12.6
	Mode I fracture energy, $G_n$ (N/m)	100
	Mode II and III fracture energy, $G_s$ and $G_t$ (N/m)	200
	Material parameter, $\eta$	2
Contact law	Friction coefficient, $\mu$	0.5

16

Table. 1 FDEM material parameters of quartz sand particles

## 1 **2.4 Validation of FDEM in simulation particle breakage behaviour**

### 2 **2.4.1 Mesh sensitivity analysis**

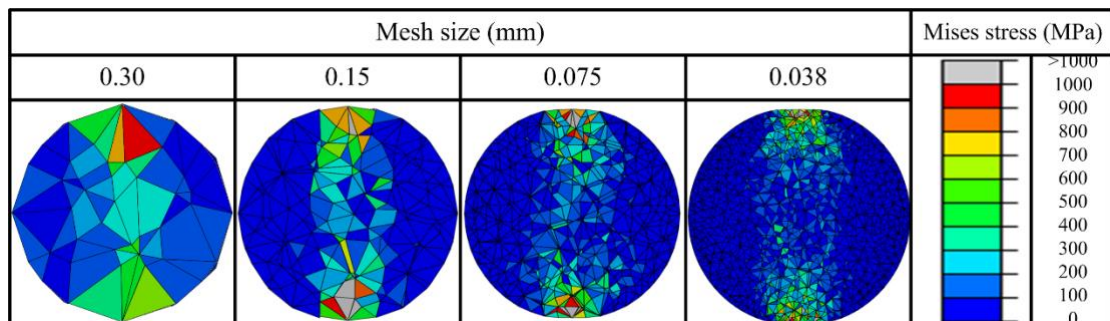
3 It has been widely known that mesh size significantly influences the mechanical  
4 behaviour in FEM-based simulation, hence the first part of study focuses on the analysis  
5 of a spherical particle with a diameter of 1.0 mm, which is close to the size of a realistic  
6 LBS particle. The material parameters are identical to the ones shown in Table. 1. The  
7 flat loading caps much harder than particles are used here for the mesh sensitivity study.  
8 To keep the quasi-static loading condition as in the experiments of [20], the loading  
9 speed in computational simulations has been checked to attain enough efficiency while  
10 keeping kinetic energy below 10% of its associated internal energy at the same time  
11 step. The upper platens ramp up to 0.2 m/s to stably introduce the contact between  
12 particle and two caps.

13 Fig. 4 illustrates the Mises stress distributions of 4 spheres just before breakage.  
14 Coarser mesh size results in less CIEs along which fracture surfaces can propagate thus  
15 generates simpler failure patterns. Finer meshes can alleviate this problem, meanwhile  
16 there is no obvious difference in stress distribution when the mesh size is sufficiently  
17 small (e.g., less than 0.075 mm).

18 The force-displacement curves of sphere simulations are shown in Fig. 5 (a),  
19 where convergence is observed with mesh size below 0.075 mm, although the  
20 0.0375 mm mesh results in slightly higher peak force and displacement at failure. It  
21 should be noted that for the smaller mesh size, the local contact area with the platens

1 increases, contributing to the higher failure stress. Meanwhile the number of fractured  
 2 CIEs increases, resulting in more potential crack paths, as compared with the courser  
 3 mesh cases. This is attributed to the possible reason why variation of the displacement  
 4 at failure is observed whilst the forces are nearly close to each other during loading  
 5 process. A mesh refinement at contact area could be helpful in future simulations. The  
 6 mesh size 0.075 mm corresponds to to 1,280 triangular surface elements, about 7,000  
 7 4-node solid elements and 20,800 6-node CIEs. To balance the simulation cost and  
 8 accuracy, 1280 triangular surface meshes are used in all following simulations. Fig. 5  
 9 (b) illustrates the evolution of kinetic energy and internal energy during loading,  
 10 inferring that the maximum ratio of kinetic energy to associated internal energy is less  
 11 than 0.1, hence the loading rate is appropriate for representing quasi-static simulations.  
 12 Notably, for platen loading on spherical particles, despite the main breakage observed,  
 13 there are still remaining parts, particularly for fine-meshed particles, that are further  
 14 compressed. Therefore, the internal energy continues to increase. Furthermore, fine  
 15 meshes can induce larger maximum internal and kinetic energies.

16

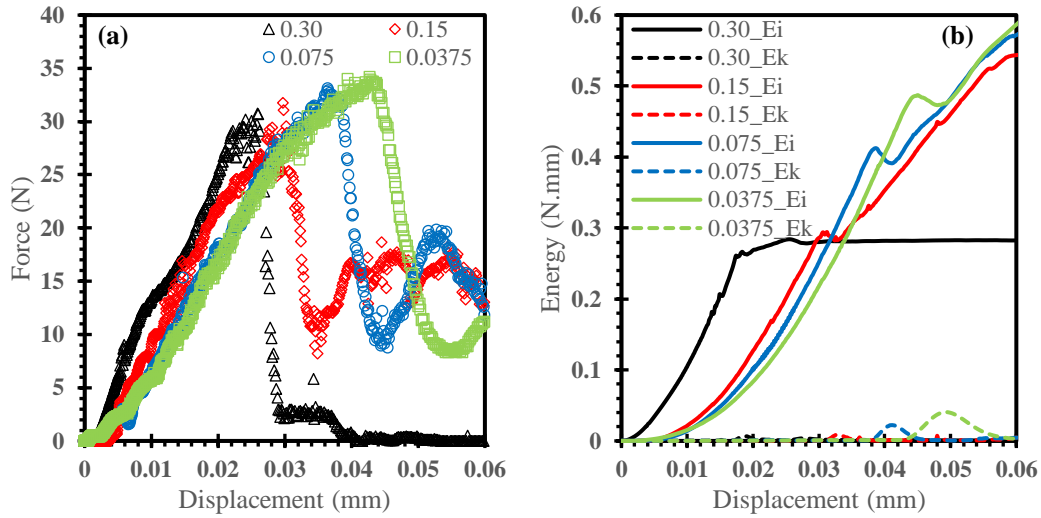


17

18 Fig. 4 The Mises stress distributions before the peak force for the spheres discretised

19 with various mesh sizes

1



2

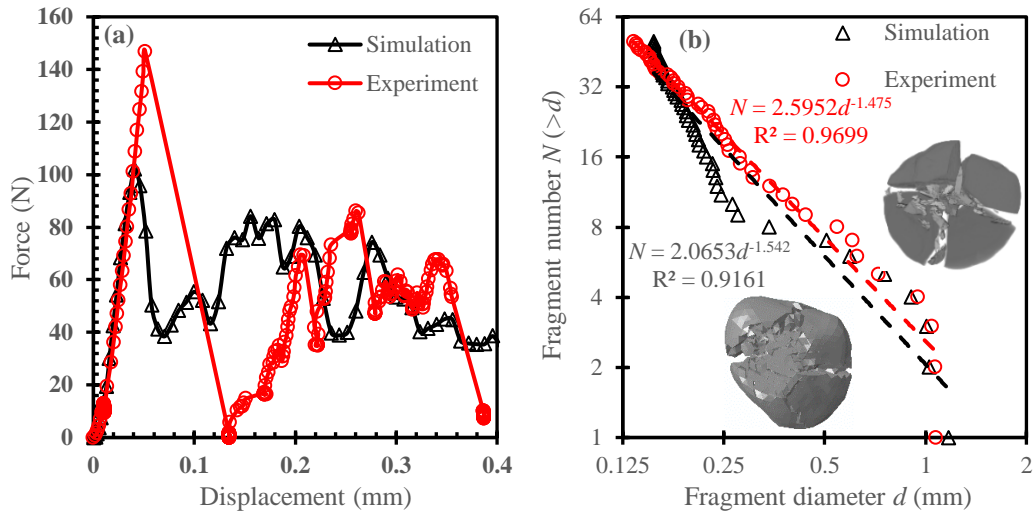
3 Fig. 5 Load-displacement curves (a) and the evolution of internal ( $E_i$ ) and kinetic  
4 energies ( $E_k$ ) (b) for the spheres discretised with different mesh sizes

#### 5 **2.4.2 Simulation of CT monitored platen loading test of a single LBS particle**

6 The model is further validated by simulating an experimental single crushing test  
7 schematically shown in Fig. 1. The implemented particle shape was taken at the first  
8 CT scan with a prestressed state of a force of around 12.8N between two platens. In the  
9 experiment, this placement of the particle can bypass redundant rotation and sliding due  
10 to complexity of contact interactions, in particular within the low loading range. Please  
11 note that the loading platens in our experiments (with a cylindrical shape of radius 5  
12 mm and height 5 mm) are made of ceramics with high Young's modulus (380 GPa) and  
13 low Poisson ratio (0.22), which is closer to rigid properties than steel (with Young's  
14 modulus and Poisson ratio about 200 GPa and 0.3) as used in other relevant studies.  
15 Furthermore, the size of loading platens is much larger than that (about 1 mm) of  
16 particle. For FEM-based simulations, the mesh size of two different contacting elastic

1 parts should have similar mesh size. If realistic platens were to be considered, a  
2 significantly heavy computational burden would be obtained. However, there is no  
3 significant difference in load-displacement curve and peak forces between cases with  
4 the rigid and elastic properties. Fig. 6 (a) shows the load-displacement curves for both  
5 simulation and experiment. Even though there are some differences regarding the peak  
6 load, which could be attributed to uncontrolled factors including differences of the  
7 platen, i.e., elastoplastic material in experiments and rigid platen used in simulations,  
8 mesh size, parameter selection, CIE model, and influence of preload on the shape of  
9 the scanned particles, it is worth noting that the curves are in a good agreement both in  
10 the elastic stage and after the first load drop. Furthermore, considering the resolution of  
11 CT images, Fig. 6 (b) plots the accumulative number  $N(>d)$  of 60 largest fragments  
12 versus their equivalent-volume diameters ( $d$ ) in log-log scales. As found in previous  
13 studies [48-49], both experimental and simulated results shown here follow fractal  
14 relationships,  $N \sim d^{-D}$ , where  $D$  is the fractal dimension. The largest 6 fragments are  
15 significantly close, which proves the ability of the model in capturing the particle  
16 breakage behaviour, even if no internal inhomogeneity was included in the simulation.  
17 Although the smaller fragment size is highly dependent on the mesh size, the fractal  
18 dimension (1.542) from simulation results is still within the typical range for sand when  
19 compared with that (1.475) of relative in-situ CT experiments. In addition, higher  
20 values in the range of  $1.77 < D < 3.06$  in FDEM studies [71-72] has been found for  
21 impact-induced single particle breakage behaviour.

22



1  
2 Fig. 6 Comparison of results for simulated and experimental 1D compression tests of  
3 single particle using flat platen: (a) Load-displacement curves; (b) Fractal distribution  
4 of fragment volume-equivalent sizes with two snapshots for the simulation (the left)  
5 and experiment (the right) at 0.135 mm displacement

### 6 3. Modelling diametral point load tests

7 In this section, traditional point load tests are applied to study the relations between  
8 particle breakage behaviour (e.g., breakage stress and fracture patterns) and particle  
9 morphology features (e.g., loading distances and curvature at contact points). The  
10 relation,  $\sigma_f = F/d^2$ , here used to calculate the breakage stress, is more suitable for point  
11 load tests and, since only pure compression forces are considered, deformation from  
12 bending moment needs to be excluded. The loading distances ( $d$ ) are the ‘diameter’  
13 connecting two load points; whereas the curvature values are quantified at the  
14 contacting vertices of the 3D surface meshes. Crushing of spherical and non-spherical  
15 particles is addressed here: spherical particles are compressed for validation of the  
16 analytical solution and studying the relation between breakage behaviour and curvature

1 at contacting areas; and CT-scanned realistic-shaped and ellipsoid particles are  
2 compressed at variable diametrical positions via rotation of the loading caps to consider  
3 not only the effects of the contact curvature, but also of load distances. Furthermore,  
4 the fracture patterns are categorised by the number of main fragments.

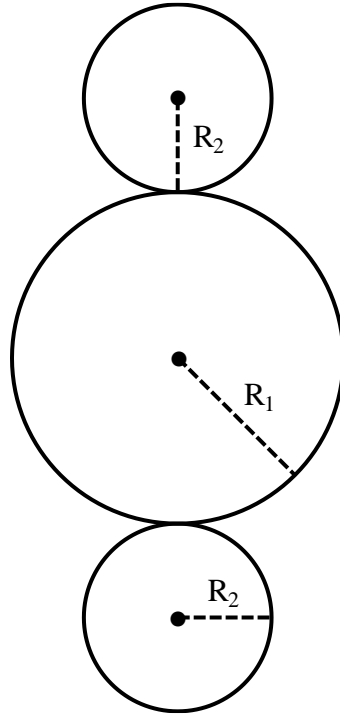
### 5 **3.1 Spherical particle breakage behaviour**

6 To assess the differences in particle breakage behaviour between point and platen  
7 loadings, the study starts with spherical particles since they have constant curvature  
8 regardless of the point considered over the surface. Three equal-mesh-sized spheres  
9 with radius 0.5 mm, 0.75 mm and 1.0 mm are compressed diametrically by two equally-  
10 sized rigid spheres as shown in Fig. 7. All material properties, as well as the friction  
11 coefficient, are taken from Table. 1. Fig 8 shows for one compressed spherical particle  
12 how its breakage or failure stress  $\sigma_f$  is significantly correlated with the relative contact  
13 radius ( $R^*$ ) in the Hertzian contact model, defined as:

$$14 \quad \sigma_f = \frac{F_{\max}}{d^2} \quad (13)$$

$$15 \quad \frac{1}{R^*} = \frac{1}{R_1} + \frac{1}{R_2} \quad (14)$$

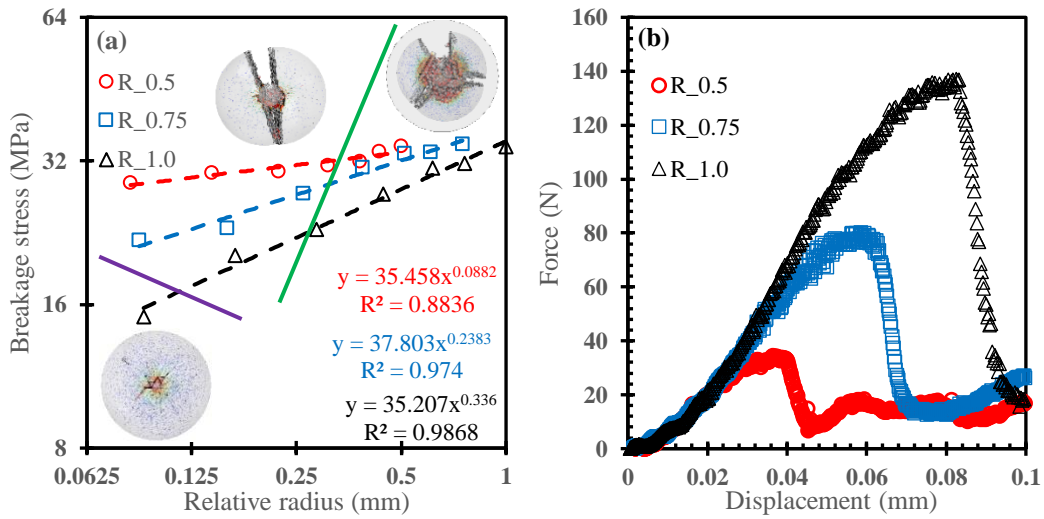
16 where  $F_{\max}$  is the maximum force applied on the particle during the compression  
17 process,  $d$  is the distance between two contact points, and  $R_1$  and  $R_2$  are the radiuses of  
18 each contacting sphere. For one sphere, there are seven simulations considering various  
19 rigid sphere radius including 0.1, 0.2, 0.4, 0.8, 1.6, 3.2 mm and  $\infty$  (i.e., flat platen), of  
20 which surfaces are always composed of 5,120 triangular elements.



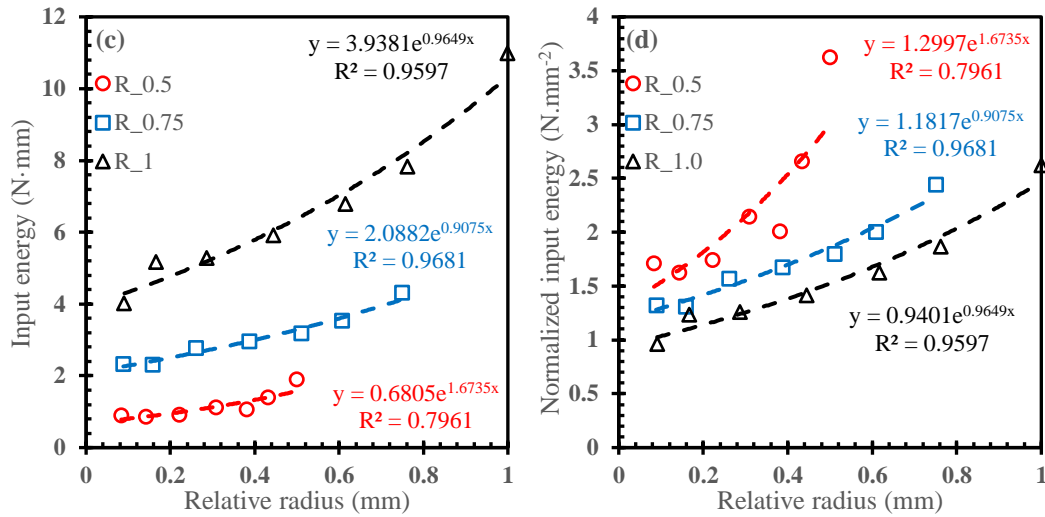
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2 Fig. 7 The 2D representation of point load tests for various sized spherical particles

3



4



1

2 Fig. 8 Breakage of spherical particles: (a) Correlation between breakage stress and  
 3 relative radius for different sized spheres (the solid lines represent approximate dividing  
 4 lines for three different failure modes: no main fragments, two main fragments and  
 5 three main fragments.), (b) load-displacement curves for platen compression, (c)  
 6 relationship between input energy and relative radius, and (d) relationship between  
 7 input energy normalized by particle volume and relative radius. (R\_0.5 means the  
 8 sphere with radius of 0.5 mm).

9

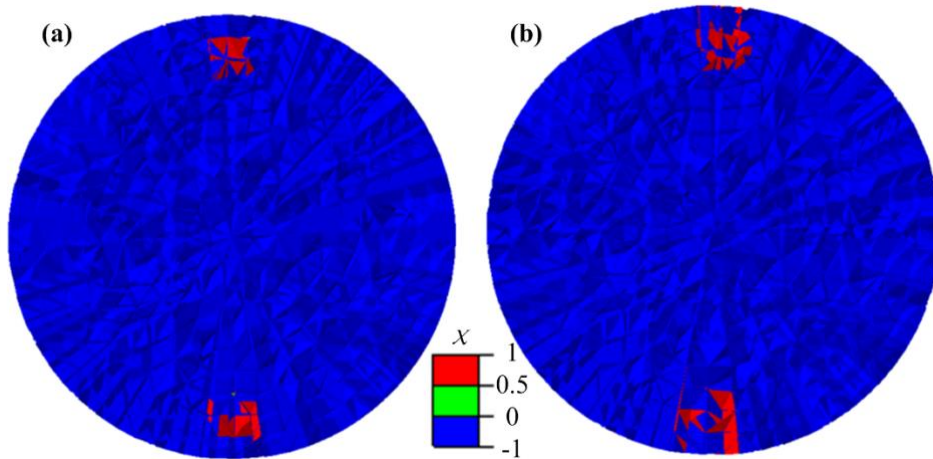
10 The linear relations between breakage stress and relative radius in the log-log scale  
 11 are observed in Fig. 8 (a), showing that the particle breakage is sensitive to the stress  
 12 concentration at the contact points. Therefore, it would be expected that the commonly  
 13 adopted maximum stress criteria should depend on the local contact curvature. Another  
 14 obvious tendency is the equivalent breakage stresses for the three sphere particles when  
 15 compressed by platens, as illustrated in Fig. 8 (b) showing no size dependency since  
 16 the simulations did not include any micro-flaw distributions. By eliminating material  
 17 inhomogeneities from the analysis, focuses can be given to the influence of the particle

1 shape alone on the breakage behaviour. As the mesh size of the three different-sized  
2 spheres is the same, for large sized particle more CIEs need to be fractured, hence  
3 requiring more energy and fracture force, as in Fig. 8 (b) and (c). With decreasing  
4 contact curvature, the spherical particle would split into more major fragments. This  
5 phenomenon is associated with the stored energy which is required at breakage to  
6 generate the fracture surfaces (Fig. 8 (c)), e.g., less force concentration and higher  
7 stored energy for cases with higher relative radii at the contact. In addition to generating  
8 two or three main fragments, when the curvature of the rigid sphere is much less than  
9 that of the breakable sphere, the small sphere would engrave into the larger sphere  
10 without major fragments, as illustrated in Fig. 8 (a). Interestingly, from the view of a  
11 deterministic model of input energy normalised by the particle volume [62], for the  
12 same particle, the volume-averaged energy is also high relevant to the relative contact  
13 radius, and the model is thus size independent, as in Fig. 8 (d).

14 As aforementioned, Russell and Wood [19] concluded that sphere breakage  
15 initiate just below the contact area, locally shear-induced and dependent on the  
16 maximum ratio of the second to the first stress tensor, denoted by a large number of  
17 small fragments underneath contact areas numerically (the fracture pattern of platen  
18 load in Fig. 8 (a)) and experimentally [20]. Even though Russell and Wood [19]  
19 conducted analytical modelling of single particle compression in the context of point  
20 loading, a uniform normal pressure is applied at contact area (see Fig. 3 of [19]), which  
21 is different from flat platen loading condition. As demonstrated by the defined mixed-  
22 mode ratio  $\chi$  in Fig. 9, it is evident that not only the maximum-stress-based crack

1 initiation but also the energy-based crack evolution result from shear deformation,  
2 which is consistent with the analytical solution.

3



4

5 Fig. 9 Early breakage types of CIEs from platen loading: (a) crack initiation; (b) crack  
6 evolution (Red denotes shear dominant crack type)

7

### 8 3.2 Realistic particle shapes and fracture patterns

9 The traditional test method for assessing the single particle strength has a pre-step  
10 of letting the particle settle down by gravity onto the platen. This results in the particle  
11 alignment towards its minimum principal dimension. However, in a granular assembly,  
12 particle orientations tend to be rearranged frequently and the resulting contact force  
13 network can be rather complex. In the following, the single particle breakage behaviour  
14 under different diametrical loading conditions is investigated. The radius of curvature  
15 at point ends, similar to [50], is 0.2 mm and the height of the loading cap is 0.1 mm  
16 above all the corresponding displacements at breakage. The particle is rotated around

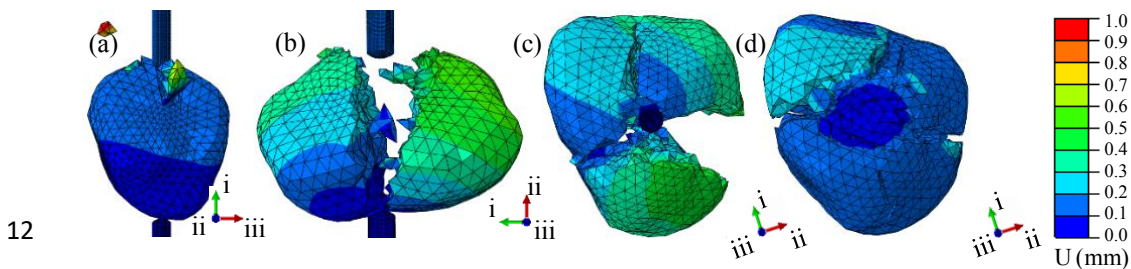
1 its centre-of-mass, as shown in Fig. 2. Using spherical harmonic reconstruction, vertex-  
2 vertex contacts, rather than vertex-face contacts, between rigid and breakable parts can  
3 be obtained before particle breakage, thus avoiding difficulties in the determination of  
4 the curvature values within flat triangles. We select 30 diametrical directions through  
5 the mass centre of the particle by equal spherical angles, including also the three  
6 principal dimensions for subsequent diametrical point loading. For comparing the  
7 breakage stress and energy between irregular and regular shaped particles, an ellipsoid  
8 with three principal dimensions equal to 2.6, 1.3 and 0.65 mm is also compressed by  
9 the same loading procedure of the LBS particle. The ellipsoid also has 1,280 triangular  
10 meshes over its surface generated by spherical harmonics.

11       Regarding the fracture patterns, there are three main types: chipping, two main  
12 fragments splitting, and three main fragments splitting. The patterns shown in Fig. 10  
13 (a), (b) and (c) are taken from loading directions aligned with the maximum, median  
14 and minimum principal dimensions, respectively, of the particle shape during the  
15 crushing processes. Notably, the direction of the minimum dimension also coincides  
16 with experimental platen loading, which is also simulated and shown in Fig. 10 (d) as  
17 a comparison. The differences between fracture patterns in Fig. 10 (c) and (d), again  
18 confirm that a reduced contact curvature avoids stress concentration, while higher  
19 energy is needed for a major breakage thus causing more severe fracture patterns. This  
20 is consistent with the results of spherical particles, shown in Fig. 8 (a), regarding  
21 breakage for point and flat platen loads.

22       No main fracture surface is found through the whole particle in the chipping mode.

1 This is because the breakage stress is relatively small, which results in less energy to  
 2 generate the fracture. By contrast, when compressed along the minimum dimension  
 3 direction, the higher breakage stress contributes to increased fracture energy producing  
 4 wider fractures and even up to three main fragments can be generated. The simulated  
 5 fracture patterns are similar to relevant experiments in [50], except for the chipping  
 6 mode. Considering that the LBS particle in their experiment is rotated in a smoothed  
 7 drum and points of long diameters usually have high curvature values, it is difficult to  
 8 compress it stably along with longer dimensions. As a result, the crushed particles are  
 9 usually fractured along smaller dimensions in [50], thus leading to higher breakage  
 10 stress similar platen loading tests and no chipping breakage mode.

11



13 Fig. 10 Three types of fracture patterns due to rotational point loading: (a) chipping; (b)  
 14 two main fragments splitting; (c) three main fragments splitting; (d) platen load towards  
 15 minimum dimension (i, ii and iii denote maximum, median and minimum principal  
 16 dimensions, respectively, of the intact LBS particle before compressed). Coloured by  
 17 the scalar displacement, U.

18

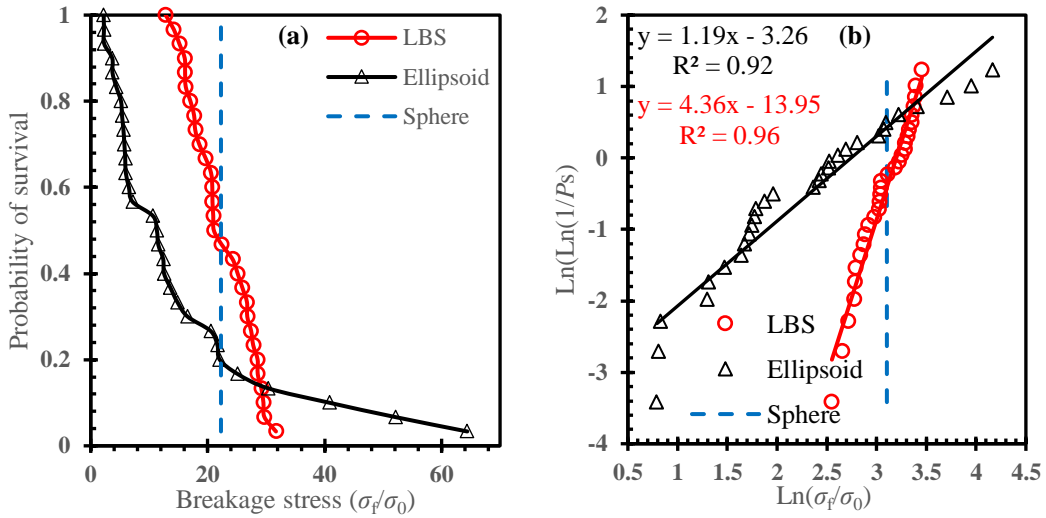
## 1 **4. Results and Discussion**

2 Here, we focus on the statistical analyses and discussion of results correlating  
3 particle morphology with breakage behaviour.

### 4 **4.1 Distributions of breakage stresses**

5 The 30 cases of diametrical directional loading presented in Section 3.2 show  
6 distinct breakage behaviour. Fig. 11 depicts the cumulative distributions (a) and the  
7 Weibull distributions (b) of breakage stresses for the LBS and the ellipsoid particles.  
8 Evidently, even though both particles have the same material properties, the breakage  
9 stresses along the different diametrical directions are contrasting. The ellipsoid particle  
10 has an averaged breakage stress (10 MPa) much lower than LBS particle (20 MPa) and  
11 a larger stress variance, due to its more elongated shape. That may be the reason why  
12 LBS particle shapes are more likely to survive in sand assemblies rather than elongated  
13 ellipsoid-like shapes [51], because their directional strength is more stable and overall  
14 the whole particle can sustain higher stress conditions when pushed against one another.  
15 Interestingly, the Weibull modulus (4.4) of the distribution for rotational breakage  
16 stresses of the LBS particle is close to the value reported in experimental platen-  
17 loading-based breakage stresses of single quartz sand particles (e.g., 4.2 in [7]). The  
18 obvious difference here is that our simulations were conducted on only one identical  
19 sand particle, while the experiments were performed on multiple particles. The nearly  
20 identical Weibull modulus also implies that particle shape may also contribute to the  
21 variance of breakage stresses in addition to internal flaws.

1



2

3 Fig. 11 Distributions of breakage stress of rotational point loading compression tests  
 4 for the LBS and the ellipsoid particles: (a) Cumulative distributions; (b) Weibull  
 5 distributions ( $\sigma_0=1\text{MPa}$ ). The dashed vertical lines indicate the corresponding values  
 6 for the equivalent-volume sphere.

7

## 8 4.2 Influences of curvature and loading distance

9 Apart from the commonly used maximum curvature ( $k_1$ ), minimum curvature ( $k_2$ ),  
 10 average curvature ( $k_M = (k_1+k_2)/2$ ) and Gaussian curvature ( $k_g = k_1*k_2$ ) from their one-  
 11 ring neighbourhood for 3D vertices [52-53], the curvedness ( $k_C = \sqrt{(k_1^2 + k_2^2)}/2$ ) is  
 12 considered here as an alternative measure from principal curvatures which avoid  
 13 negative values found in  $k_1$ ,  $k_2$ ,  $k_M$  and  $k_g$ , and seamlessly combines two contact points.  
 14 The combined curvedness is defined in similar manner to that of curvedness ( $k_C$ ):  $k_{com}$   
 15  $= \sqrt{(k_{C1}^2 + k_{C2}^2)}/2$ , where  $k_{C1}$  and  $k_{C2}$  are curvedness of two contact points.

16 Fig. 12 shows that the particle breakage stress is highly correlated with both

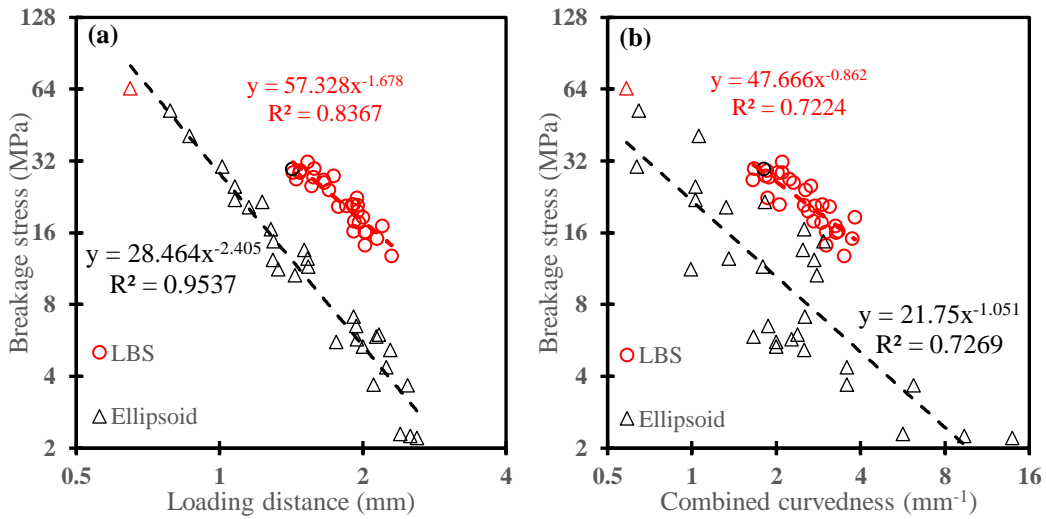
1 contact distances and the combined curvedness of two contact points, for both LBS and  
2 the ellipsoid particles. However, the ‘diameters’ would have some implicit correlation  
3 with the curvature values of their two end points for a given shape. Fig. 13 shows a  
4 general monotonic relationship between the diametrical length and corresponding  
5 normalised combined curvedness sweeping through the particle surfaces. Fig. 14 shows  
6 a 3D plot for breakage stress of rotational point loading compression, versus loading  
7 distance and its combined curvedness. It has been widely accepted that particle strength  
8 is size dependent and the correlation in log-log scales is linear [7]. For simplicity, the  
9 load distances ( $d$ ) between two platens just before compression is usually considered as  
10 the particle sizes. From the FDEM simulations using rotational point loading tests, the  
11 breakage stresses are found to be linearly correlated with  $d$  in log-log scale for the same  
12 particle. Thereby, it seems that the so-called size effect may be due to variations of  
13 loading distances rather than the actual size.

14 Furthermore, the breakage stresses of point loading close to the minimum principal  
15 dimension directions are pointed out in Fig. 12 and among the highest values. In some  
16 cases, higher breakage stresses may be observed since another major influencing factor  
17 is the combined curvedness of the two diametrical contact points. Consequence of the  
18 above conclusion, traditional compression tests using platens somewhat overestimate  
19 the breakage stresses.

20 In the perspective of sand particle samples in continuum scale, their anisotropic  
21 fabric features are usually quantified by fabric tensors, and a popular directional fabric  
22 tensor is defined and based on the distribution of the principal dimension directions

1 [54]. Meanwhile, deformation-induced inhomogeneity of granular materials is also  
 2 observed during the loading [55-56], also called yielding [57]. Therefore, the fracture  
 3 criteria based on the shortest dimensions is not sufficient to assess the breakage  
 4 behaviour of irregular particles within a granular assembly. At the grain scale, particle  
 5 strengths depend on the contact direction, i.e., stress and fabric tensors, can be used to  
 6 better describe the continuum breakage behaviour.

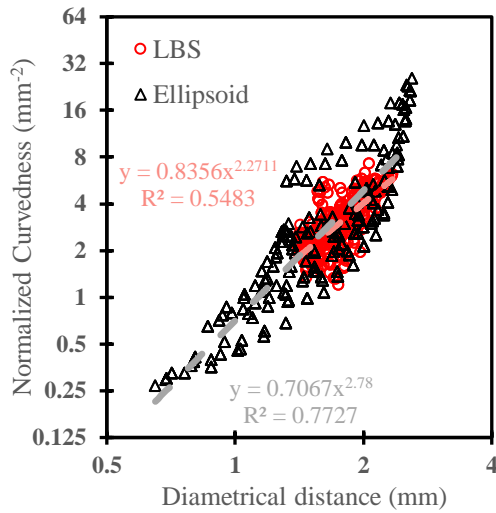
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8

9 Fig. 12 Breakage stresses vs (a) loading distances and (b) combined curvedness for  
 10 rotational diametrical point loading for the scanned LBS particle and an ellipsoid. (The  
 11 corresponding red triangle and black circle represent point-loading tests towards the  
 12 minimum dimensional directions for the ellipsoid and LBS particles.)

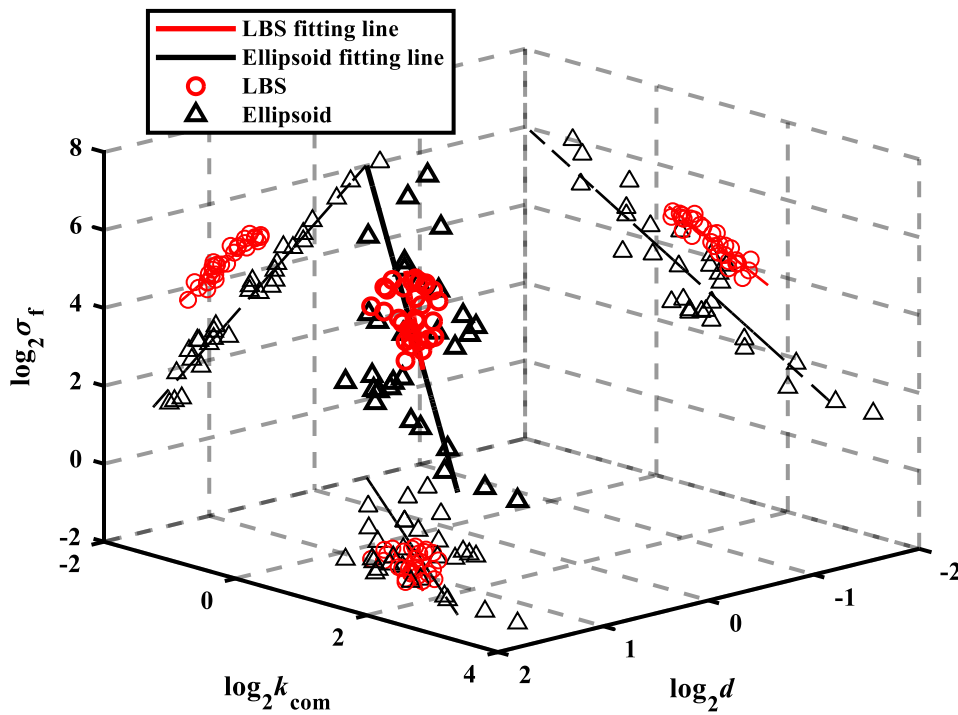
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1

2 Fig. 13 Distributions of normalised curvedness multiplied by its corresponding  
 3 diametrical distance versus 321 diametrical lengths crossing the mass centre for the  
 4 LBS and ellipsoid particles

5



6

7 Fig. 14 Unique point-load-induced single particle breakage stress curve in logarithmic  
 8 scale  $\sigma_f$ - $k_{com}$ - $d$  space

### 1 4.3 Energetic analysis of particle breakage

2 In geomechanics, energy evolution is the main concern of constitutive modelling  
3 in continuum scale [58-64]. In this section, the energy evolution during the breakage  
4 processes is analysed at the grain scale. In single particle crushing experiments two  
5 energy forms, namely input energy and fracture energy, can be retrieved. Input energy  
6 is calculated from load-displacement curves, while fracture energy is obtained from the  
7 newly created fracture surface areas, which are directly obtained from CT images [65-  
8 66]. Fig. 15 provides (a) input energy and (b) increased surface area of the experimental  
9 and simulated platen-loading of single particle breakage. During the severe breakage  
10 process, however, fragments may be too small for the resolution of CT images to  
11 accurately capture the total surface area. In addition, bigger internal cracks or flaws can  
12 also reduce accuracy of calculating surface area. While in FDEM simulations the  
13 cracked area is highly dependent on pre-existing CIEs, the size of which determines the  
14 smallest available fragments. That is the reason why discrepancies in terms of fracture  
15 area are found between experimental and simulation results, in particular for Scan 3 in  
16 the CT monitored experiments, shown in Fig. 15 (b). In contrast, for Scan 2 fewer  
17 fragments are generated and thus less affected by CT resolution in experiments and CIE  
18 size is found, demonstrating the capability of FDEM simulations.

19 In Fig. 16, four types of energy are analysed, namely maximum elastic strain  
20 energy ( $E_E$ ), input energy by external works ( $E_W$ ), damage energy ( $E_D$ ), and friction  
21 energy ( $E_F$ ) at the final stage with no load bearing capacity. In this study, the damage  
22 in CIEs is defined through (in Fig. 3):

$$1 \quad \boldsymbol{\sigma}^c = (1 - D') \boldsymbol{\sigma}^u, \quad (15)$$

2 where  $\boldsymbol{\sigma}^u$  is the undamaged stress and  $D' \in [0,1]$  is the damage parameter. Thus the  
3 strain energy is

$$4 \quad E_S = \int_0^t \left( \int_V (1 - D') \boldsymbol{\sigma}^u : \dot{\boldsymbol{\varepsilon}}^{el} dV \right) dt. \quad (16)$$

5 Simultaneously, the damage  $D'$  is time dependent and the strain energy ( $E_S$ ) could be  
6 decomposed into the recoverable elastic energy ( $E_E$ ) and damage-dissipated energy  
7 ( $E_D$ ):

$$8 \quad E_E = \int_0^t \left( \int_V (1 - D'_t) \boldsymbol{\sigma}^u : \dot{\boldsymbol{\varepsilon}}^{el} dV \right) d\tau = \int_0^t \left( \int_V \frac{(1 - D'_t)}{(1 - D')} \boldsymbol{\sigma}^c : \dot{\boldsymbol{\varepsilon}}^{el} dV \right) dt, \quad (17)$$

$$9 \quad E_D = \int_0^t \left( \int_V (D'_t - D') \boldsymbol{\sigma}^u : \dot{\boldsymbol{\varepsilon}}^{el} dV \right) d\tau = \int_0^t \left( \int_V \frac{(D'_t - D')}{(1 - D')} \boldsymbol{\sigma}^c : \dot{\boldsymbol{\varepsilon}}^{el} dV \right) dt. \quad (18)$$

10 For explicit definition of friction energy ( $E_F$ ), the surface traction vector ( $\mathbf{t}$ ) can be split  
11 into surface normal load ( $\mathbf{t}^l$ ) and frictional traction ( $\mathbf{t}^f$ ) along boundary  $S$ . Then it can  
12 be written as

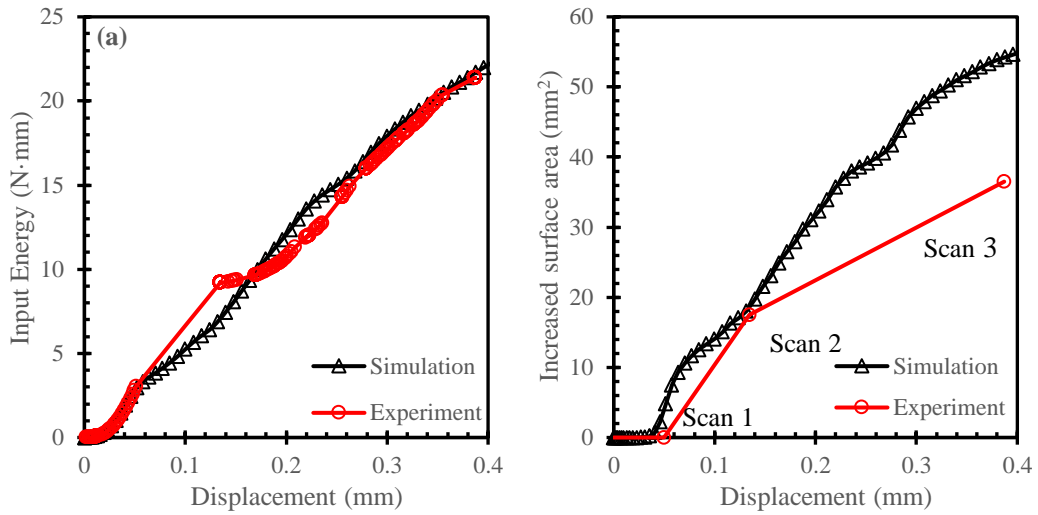
$$13 \quad E_F = \int_0^t \left( \int_S \mathbf{v} \cdot \mathbf{t}^f dS \right) dt. \quad (19)$$

14 where  $\mathbf{v}$  is the velocity field vector.

15 Except for the friction energy, all other 3 energy terms conform to Weibull  
16 distributions well in Fig. 16. Interestingly, regardless of the particle LBS particle or  
17 ellipsoid, the Weibull modulus of maximum elastic strain energy is evidently higher  
18 than those of other energy distributions; that is, with respect to the loading directions,

1 the maximum stored strain energy changes less than other energy types. Although the  
2 Weibull modulus of the maximum elastic strain energy for the LBS particle (7.3298) is slightly  
3 lower than that of the input energy (7.4427), the two values for the ellipsoid particle are 17.278  
4 and 3.5808, respectively. The less variance in maximum elastic strain energy than the  
5 breakage stress implies that the stored maximum elastic strain energy can be a  
6 replacement as a breakage criterion. In particular, for regular-shaped particles, of which  
7 maximum elastic energy can be obtained analytically by integrating the Hertzian  
8 contact force for assessing the particle breakage [67-68]. We further compare the  
9 maximum  $E_E$  and  $E_D$  during the compression process of rotational point tests for both  
10 LBS and ellipsoid particles in Fig. 17, where the maximum  $E_E$  is significantly close to  
11  $E_D$  for the 60 tests. Although in this study diametrical point load with less sliding or  
12 rotation in fracturing processes is applied, friction dissipation yet takes more parts of  
13 input energy than any other energy components, as illustrated in Table. 2. The sliding  
14 between fractured parts also contributes to the total friction energy. In continuum-scaled  
15 simulation of granular samples considering particle breakage [69-70], the ratio of  
16 friction dissipation to input energies is larger than that of point-load-induced single  
17 particle breakage due to the sliding of particles.

18

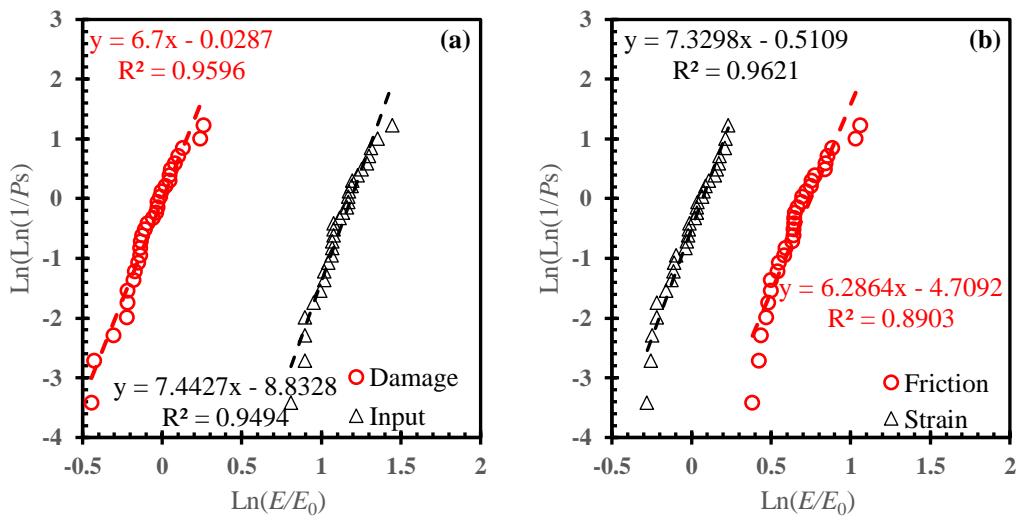


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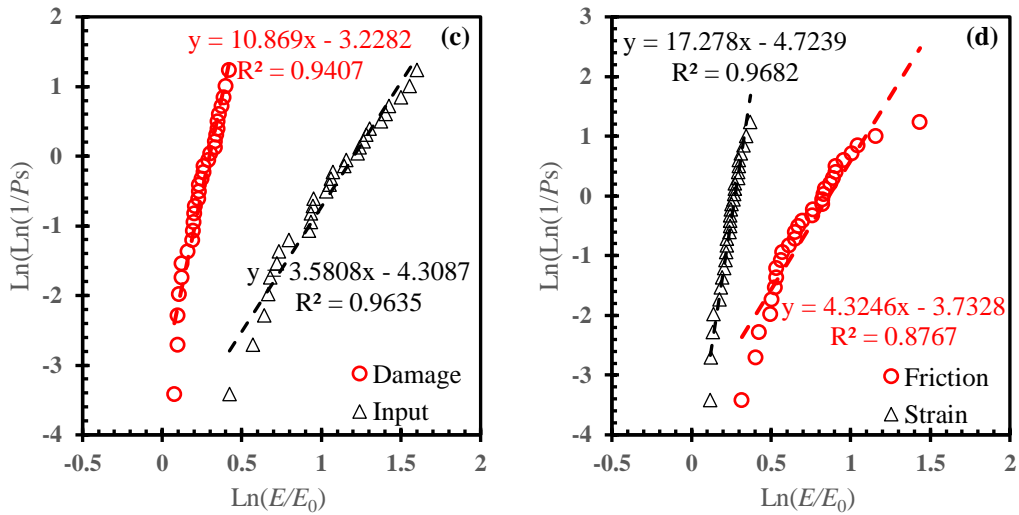
2 Fig. 15 (a) Correlation between input energy and (b) cumulatively increased surface

3 area and displacement for experimental and simulated platen load tests in Fig. 1

4

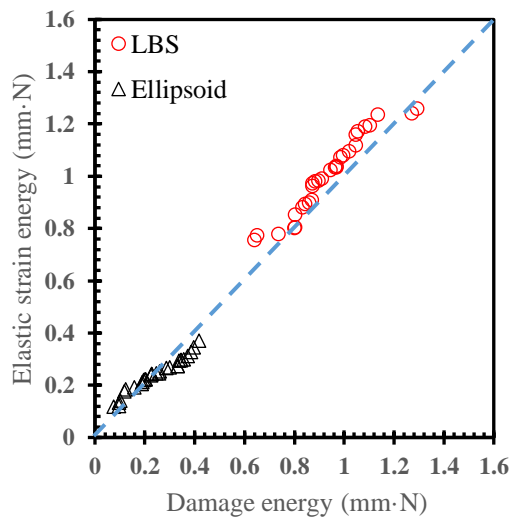


5



1  
2 Fig. 16 Distributions of energy terms for rotational point load: Compression of LBS ((a)  
3 and (b)) and ellipsoid ((c) and (d)) particles ( $E_0=1\text{N}\cdot\text{mm}$ )

4



5  
6 Fig. 17 The correlation between the maximum  $E_E$  and ultimate  $E_D$  for rotational point  
7 compression tests for the LBS and ellipsoid particles

8

Particle kind	$(E_E/E_W)$	$(E_D/E_W)$	$(E_F/E_W)$
LBS	0.3288	0.3071	0.6384
Ellipsoid	0.2378	0.2394	0.6918

1 Table. 2 Summary of average ratios of maximum  $E_E$ ,  $E_D$  and  $E_F$  to  $E_W$  for LBS and  
2 ellipsoid particles

### 3 **5. Conclusions**

4 In this study, a combined FDEM approach has been adopted to simulate particle  
5 breakage behaviour emphasising on the particle shape effects. A CT-monitored  
6 experiment of a single particle crushing was taken as a benchmark and the analytical  
7 solution was verified. The main conclusions withdrawn from this study are as follows:

- 8 • The simulations of rotational point loading provide a powerful tool to assess the  
9 same non-spherical particle shapes with multiple breakage measurements that  
10 are impossible to study experimentally.
- 11 • Diametric-load-induced particle breakage behaviour, in terms of breakage stress  
12 and fracture patterns, is governed by both their loading distances and curvature  
13 at contact points. Therefore, the new breakage stress concept proposed based on  
14 load distance and curvedness at contact points provides an important reference  
15 for future numerical researches on single particle breakage.
- 16 • The maximum elastic strain energy during deformation was found to be an  
17 effective energy breakage criterion to determine strength of brittle particles.

18 These conclusions are generalised from three types of shapes investigated in this  
19 study, i.e., spheres of difference sizes, ellipsoids, and LBS particles, of which global  
20 morphology features are isotropic, symmetric and irregular, respectively. In a  
21 continuum-scaled particulate environment, since most fractured particles are in contact  
22 with more than two neighbour particles, it is necessary to consider the contribution of

1 all forces to the stored elastic strain energy in them. Care should be taken when taking  
2 critical elastic strain energy in the strength model as the sum of the strain energies of  
3 all contacts of a particle as a hypothesis and until experimental justification becomes  
4 possible. The results and findings in this study provide a promising way to unravel the  
5 dominating fracture mechanisms for irregular particles, and a conceptual numerical  
6 framework for particle breakage. It also enables future investigations to understand the  
7 behaviour of brittle granular media, and offers a new way to interpret the available  
8 experimental data regarding particle fracture behaviour.

9

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